

Investigation of the ceramic tiles that contains boron frit

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The floor tiles contain porosity depending on the production process. Therefore, water absorption is higher for the floor tiles. The water absorption permeability leads to problems such as visible spots on the surface of the products after laying. This situation returns to firms in the customer complaint. Impervious engobe has been investigated as a possible solution to this problem. In this study, a more sintered engobe was tried to be obtained. The amount of glassy phase increased due to increased boron frit, so water permeability decreases were observed. The permeability problem has been occurred in the samples which have prescription rates below 20% boron frit. XRD, SEM, DTA, Thermal Microscope and expansion analysis was performed. According to the results obtained from SEM images of impervious engobe samples were found to be more compact. This shows that spaces have been closed due to increasing amount of the boron frit. The lower sphere temperature have observed in the thermal microscope analysis for the engobes which contain more boron frit. So the sintering temperature has been reduced and thus the floor tiles permeability has been reduced.

Keywords: Engobe, Boron frit, Permeability.

Introduction

Engobe is using to concealed the ceramic body colour by covering the product between body and the glaze with the required colour and optimum density. Also it is using to fit the thermal expansion between the body and the glaze. The other role of the engobe is reducing water absorption [1]. The engobe layer is situated between the tile body and glaze and acts as an obscuration layer. It has been shown Fig. 1.

Boron, as boric oxide is an important chemical component of many raw materials and a variety of frits used as intermediate products by the ceramic industry [2].

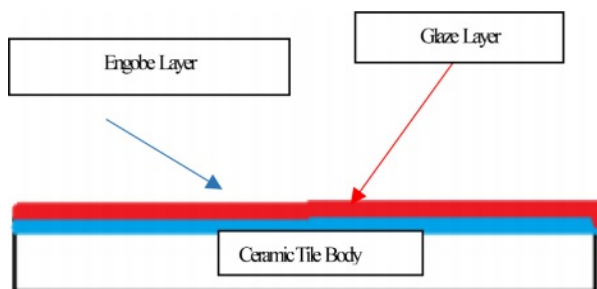


Fig. 1. The cross-section of the engobe layer.

Boron is an important component of glaze and engobe compositions. By the development of the technology, boron had been used in glazes and engobes by inclusion in a frit to render it insoluble. Boron had been many beneficial functions in ceramic glazes by using as a frit. Boron is a flux that never increases thermal expansion and also it improves chemical durability. We can eliminate hazardous lead oxide in engobes and glazes by using boron oxide. Also feldspars are included in ceramic body as a fluxing agent [3]. Fluxes are the materials which lower the melting point of a glaze. They can be called melters. Silica melts by itself but at a very high temperature. Therefore it needs additions of flux to make a practical glaze. The most common flux for temperatures below 1100 °C is lead oxide (PbO), but since it is poisonous it is no longer used in modern crockery glazes. Another powerful flux is boron or boric oxide, B₂O₃, which is not poisonous and is used in glazes in the form of borax or boric acid [4].

In general, the higher the boric oxide content, the lower the firing temperature of the glaze. In ceramic tiles, boric oxide remains an essential component in almost all types of tile, whether wall, floor, porous. Frits are used in engobes to render the desired soluble elements sodium, potassium and boron insoluble in water [4]. They need to be insoluble since soluble elements will migrate during drying processes that give rise to glaze and engobes defects as well as the possibility of effluent problems as they would be present in wastewaters. The frit also ensures that the melting process commences at an early stage, that is before the glaze firing process itself. This

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ensures that high gloss in the engobes and the glaze firing process is easily obtained.

Floor tiles contain porosity depending on their production process, raw materials and fields of application in their structures. Consequently, water absorption of floor tiles is high. This water absorption causes stain problems on the surface of the product because of water permeability after covering. This situation gets back as customer complaints to the companies. For the solution of the problem impervious engobe studies have been carried out [5].

It was observed in E. Günay's research that, the sintering time was less effective in the sintering behavior of the system compared to the sintering temperature and B_2O_3 additions [6].

$ZrSiO_4$ is widely used as a glaze opacifier in the ceramic glaze industry, and many studies have reported the effect of frit composition on the solubility of zirconium compounds [7, 8]. Opacity results from the difference between the refractive index of an opacifier and a glassy matrix [9]. Thus, the greater the difference in refractive indices between these two, the higher the light scattering and hence the more satisfactory the opacity [10]. The emitted light intensity also depends on the particle size of the crystalline phases, as long as it is larger than the wavelength of the light. The finer the particle size, the larger the surface area, the greater the number of reflective surfaces, and the more frequently the light's path is disrupted [11]. This produces a greater reflection and higher opacity. Another factor that affects the opacity of a material is the amount of opacifying phase present. Increasing the number of crystallites also increases the number of reflective surfaces, which increases the opacity of the material. The opacity of the glaze may result from the nature of the glaze or the presence of opacifying agents in the glaze. 23 Among commercial frits, those containing zircon ($ZrSiO_4$) and zirconia (ZrO_2) attract great attention. $ZrSiO_4$ causes the formation of opaque frits often referred to as "zirconium white" [12]. Also frit systems were developed for fast single firing floor tiles [13].

The purpose of this study is developing new floor tile engobe compositions that have maximum whiteness and maximum opacity level in the course of improving floor tile engobe characteristics. In this sense, it is aimed to decrease porosities by increasing the amount of boron frit. So that vitreous phase has been increased and water absorption has been decreased. For that purpose imperviousness test has been performed by applying different engobes which have been prepared by using boron frit on floor tile structures to pull down sintering temperature by increasing melting phase.

Experimental Methods

The experimental studies for the purpose which is indicated above have been completed in three stages.

In the first stage, it has been tried to decrease melting phase for the highly opaque engobe compositions. In the second stage perviousness tests after sintering and L, a, b measurements of the same engobes have been performed. In the third stage characterization processes of the engobes have been completed. Analysis has been made by receiving the images of XRD, thermal microscope, dilatometer analysis and SEM. By this means, the sintering mechanism and the arising phases have been scrutinized. 20 different types of engobe compositions which were prepared in the first stage were grinded in 100 gr. capacity of dry matter, lab-type alumina ball porcelain mill in 10 minutes. Litre weight measurement was made at the end of grinding for all the engobe compositions. After determining the tare of the pycnometer, the engobe is loaded and weighed. The value which was seen while weighing is multiplied by 10 and the result gives the litre weight in gr/lit. approximately 1430 g/lit weighed engobes which are used in operating conditions were prepared for the floor tile engobes. Application has been performed with airbrush. 35 gr engobe was applied for a 33×33 cm tile.

Chemical analysis of fired engob samples was performed using the Spectro Brand X-lab 2000 device.

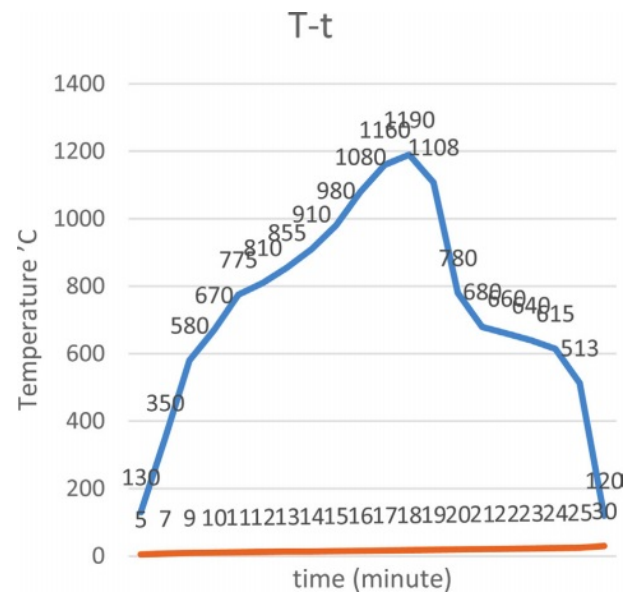


Fig. 2. The firing cycle of the engobes.

Table 1. The amount of raw materials of the engobe prescriptions.

Raw Material	Amount %
CaCO ₃	16
Borik Acid	16,0
Alumina	2,0
Quartz	24,0
Zirkon	9,0
Albit	10,00
Total	100

Table 2. Chemical analysis of fired engob samples.

Raw materials%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Frit	0,02	58,5	10,59	0,11	0,08	11,52	0,02	3,32	0,33	0,02	0,01	6,25	9,23
Std.	0,06	54,85	28,76	0,46	0,36	2,54	0,35	2,84	0,80	0,61	0,04	7,63	0,16
E6	0,25	57,98	23,16	0,40	0,35	3,46	0,3	2,13	0,93	0,10	0,01	10,05	0,84
E7	0,05	55,87	11,87	0,24	0,15	7,15	1,28	1,37	2,25	0,04	5,88	8,83	5,01
E10	0,12	57,98	21,15	0,45	0,28	2,32	0,02	2,29	1,00	0,05	0,01	6,92	0,77
E11	0,20	57,68	19,39	0,44	0,38	2,53	0,02	2,24	1,06	0,03	0,006	11,27	4,75

They were sintered in the kiln for 30 minutes on 1190' C after drying 10 minutes by using the standart glaze. It has been given at Fig. 2.

It has been given in Table 2 amount of raw materials which belong to frit prescription and in Table 1 comparative amount of raw materials which belong to each engobe prescription has been given. It is very important for this study the amount of B₂O₃.

Results and Discussions

Permeability Test Measurements

The floor tiles engobe perviousness tests were done in business laboratory of Kütahya Ceramic Factory. After being clinged to a grouted plate, the floor tiles waited for 1 day. Than it is as certain that whether they left stain or not by inspection.

According to the test results it is determined that E7, E8 and E9 engobes are impervious. In Table 3. But whitness of E9 engobe which was developed without zircon is very low Fig. 3. E7 and E8 engobes are greyer than the standard engobe. The studies continue in order to approximate the whitness of the engobes to the standard engobe. Whiteness value of E10 engobe is very close to the standard engobe but perviousness has been observed. The perviousness E11 engobe is zero. However, greyness is a matter as it can be understood by looking at L, a, b values. Although E12 engobe does not have a perviousness problem, it has got grayness issue. Despite the whitness value of E5 engobe is quite good, it has got a perviousness problem.

E13, E14, E15 engobes do not have perviousness problems but it is understood that they are hard engobes. Although E16, E17, E20 engobes do not have perviousness problem, greyness issue has been observed by inspection

Colour Measurements

Colour measurement had been performed with Minolta CR 300 measuring device. A colour value-detector which detects colour value of the device was located on the surfaces of scoured engobes. Colour values were observed from the indicator of the device after being pushed the button. The results has been shown at Fig. 3.

In previous studies, various attempts were made to

determine a relationship between engob opacity and zirconium silicate. For this purpose, an industrial engob containing standard zirconium silicate (A) and another engobe without zircon (B) were tested. After these two engobes were applied to the body, they were fired at different temperatures. It was observed that the engob layer without zircon had a lower whitness index at all tested temperatures. For example, at 1140 °C, engob A behaved especially at L and Ib (whitness index) values of 90.05 and 77.5. However, engob B without zircon gave whitness values of 78.02 and 72.2 in the same parameters [14].

According to this result, it is seen that between the zircon ratio and opacity, the effect of zircon on the mechanism increases. As a general view, engob opacity with additional zircon may cause different mechanisms.

Table 3. The perviousness test results.

Prescription	Permeability
Std engobe	permeable
E1	permeable
E2	permeable
E3	a little
E4	a little
E5	permeable
E6	impermeable
E7	impermeable
E8	impermeable
E9	impermeable
E10	permeable
E11	impermeable
E12	impermeable
E13	impermeable(hard)
E14	impermeable(hard)
E15	impermeable(hard)
E16	impermeable
E17	impermeable
E18	impermeable
E19	permeable
E20	Impermeable

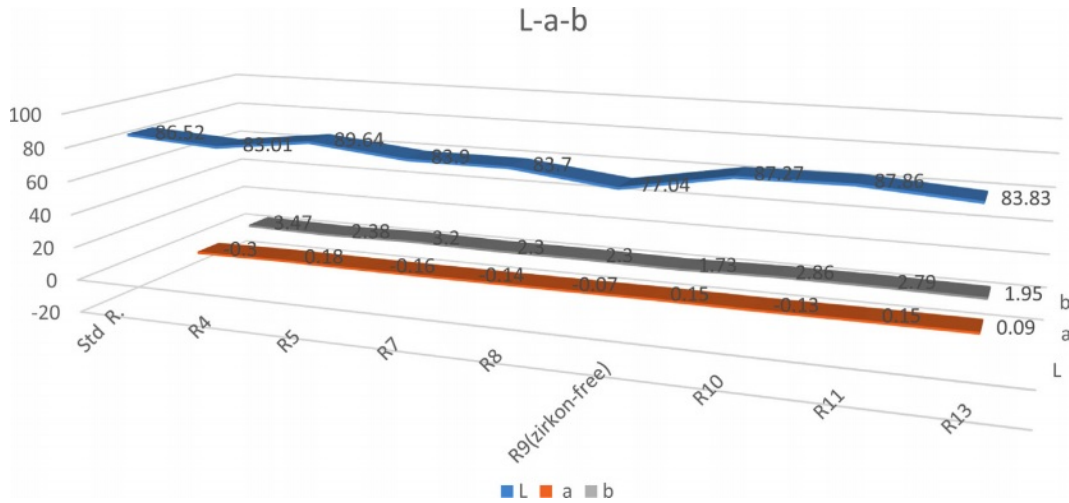


Fig. 3. The results of the colour measurements.

If excess zirconium silicate particles are added to the system, the particles may become insoluble in the glassy phase during firing and then begin to separate into the glassy phase. As the zirconium particles begin to solidify again during the opacification in the cooling cycle, the zirconium silicate becomes insoluble in the glassy phase [14].

All the colours can be described in 3 coordinate system.

- Coordinate X, L = white - black
- Coordinate Y, a = red - green
- Coordinate Z, b = yellow - blue

The degree of opacity depends on other engob components such as aluminum and zinc. X-ray studies have shown that the main opacifying phase in cooked engobes is zircon. When used with frit, the opacifying phase was determined to be zircon as a result of the heat treatment, as seen in Fig. 4 [15].

ZrSiO₄ (Zircon) has been widely used as glaze opacifier in the ceramic glaze industry and several researches have reported the influence of frit composition on the solubility of zirconium compounds. Zircon gives rise to opaque frits which are glossy, viscous and with a low fusibility [13, 15].

X-Ray Diffraction Analysis

This analysis was performed with X-ray diffraction device by using the Rigaku Miniflex series. CuK α radiation which was acquired by applying 40 kV voltage and 30 mA current to the Cu tube was performed by using ($\lambda=1,54046 \text{ \AA}$).

Qualitative stage analysis was performed with XRD to locate the kind of crystal stages that consisted on engobes after sintering, investigate crystal progress that occurred as a result of sintering at the same heat and compare the crystal amount of the engobes relatively.

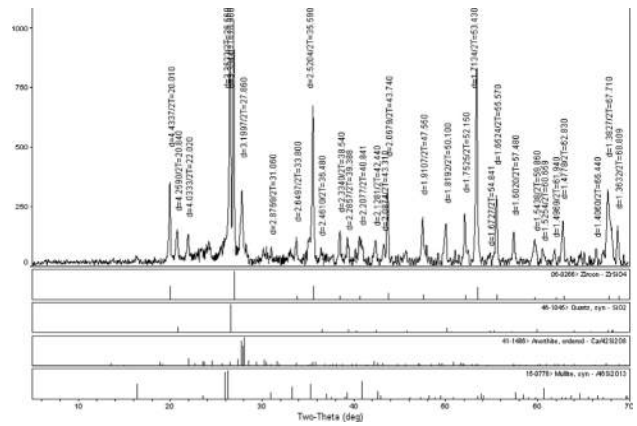


Fig. 4. Xrd pattern of the fired standart engobe.

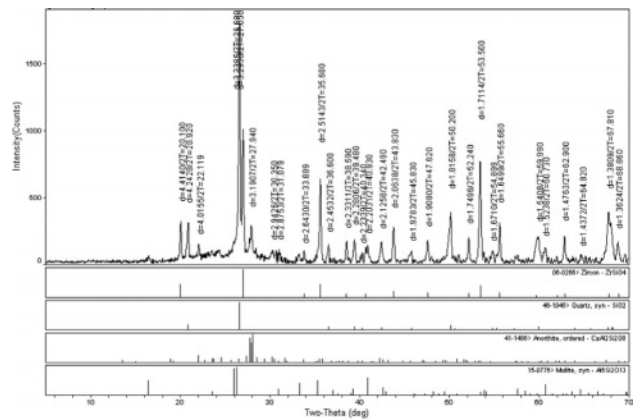


Fig. 5. Xrd pattern of the fired E11 engobe.

That has been shown in Fig. 4 and Fig. 5.

According to the XRD analysis of scoured engobe zircon stage has developed as whitening stage (Fig. 4) According to the XRD analysis of the fired samples, it was observed that the zircon phase was formed as the

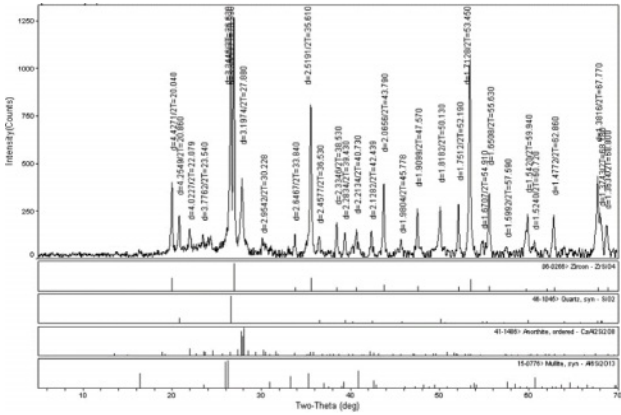


Fig. 6. Xrd pattern of the fired E6 engobe.

whitening phase.

Scanning Electron Microscopy

SEM images have been performed in the SAM laboratory.

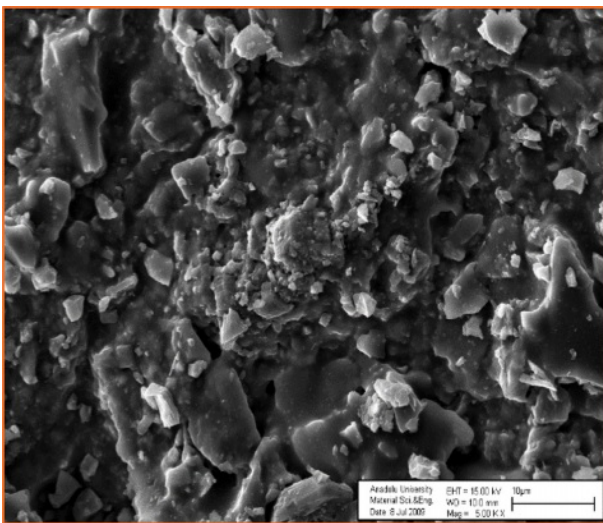


Fig. 7. SEM image of E11 Engobe.

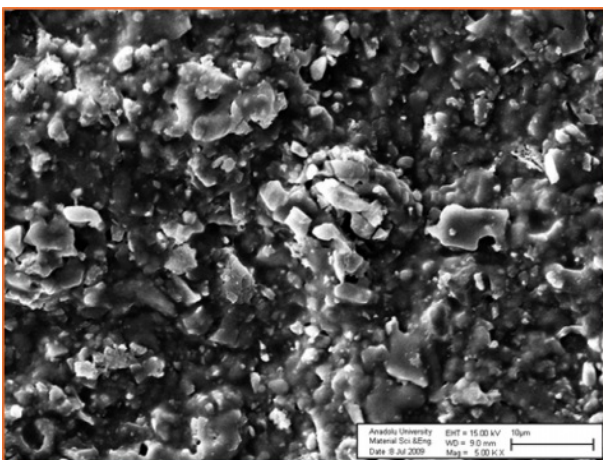


Fig. 8. SEM image of standart Engobe (5000 X).

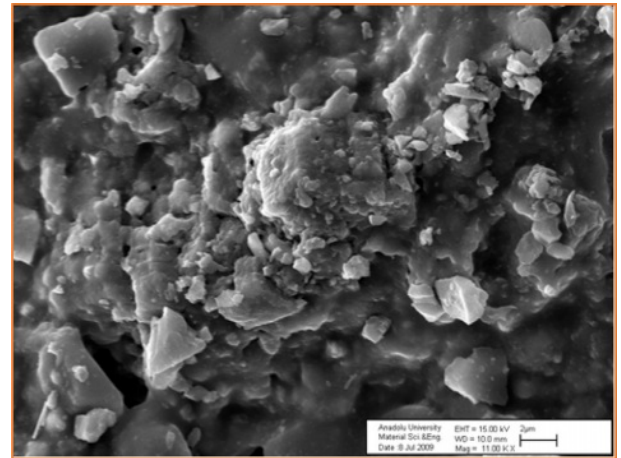


Fig. 9. SEM image of standart engobe (11000X).

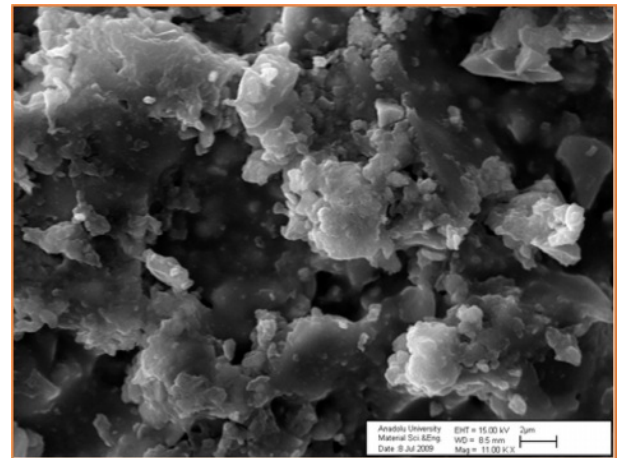


Fig. 10. SEM image of E11 engobe (11000X).

E11 engobe is more compact than the standart engobe. It has been shown in Fig. 6 and Fig. 7. Thus, we can understand that the permeability problem can be solved with more compact engobe layer.

Also it has been shown more detailed in Fig. 8 and

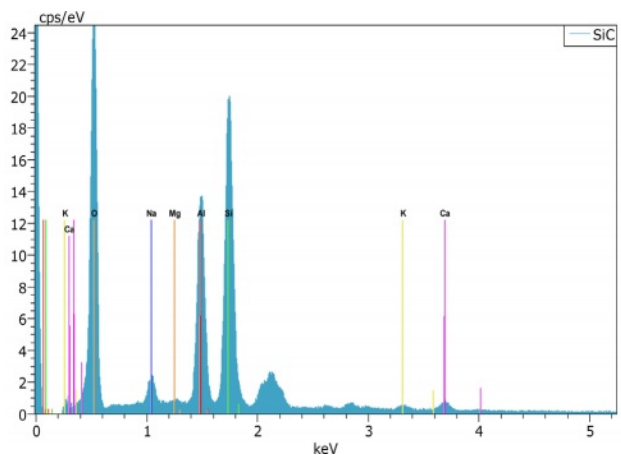


Fig. 11. Edx Analysis Image of E11 Engobe.

Fig. 9 with 11000X magnification.

In accordance with the results which have been obtained from SEM images, it is observed that E11 engobe is intenser than the standard engobe. This shows that the spaces among crystals close better with increasing of vitreous phase. As a result, water absorption has decreased and the perviousness problem has been removed in E11 engobe. Primary mullite formation has been observed in performed in SEM images. SEM images support XRD results. Primary mullite formation has maintained high chemical resistance, low thermal expansion and good thermal shock resistance to the structure (Figs. 7-10).

Differential Thermal Analysis

Differential Thermal Analysis is performed by Netzsch STA 409EP model device in Dumlupınar University.

According to Fig. 12

- Body water; Separated at 68 °C-100 °C
- Organic degradation; at 315 °C
- Removal of crystal water from kaolin; at 506 °C
- Quartz transformation; at 574 °C
- 1. mullite formation; at 1006 °C

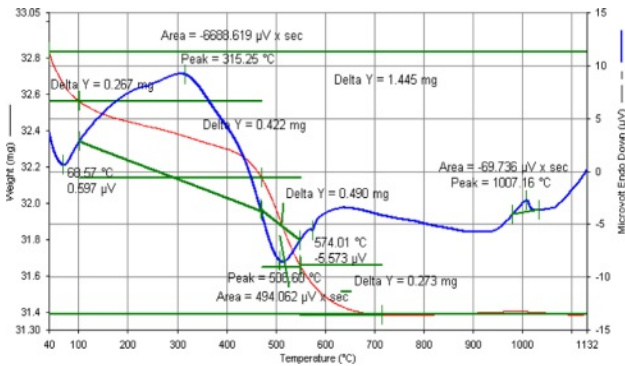


Fig. 12. E11 engobe DTA.

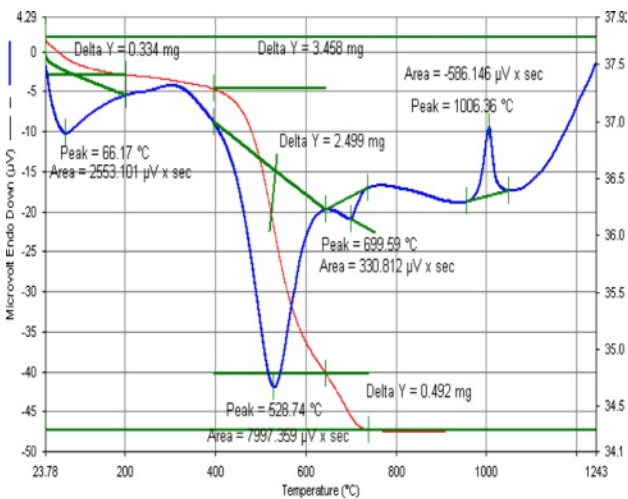


Fig. 13. E5 engobe DTA.

Thermal Microscopy Analysis and Thermal Expansion Analysis

Thermal microscopy analysis has been performed

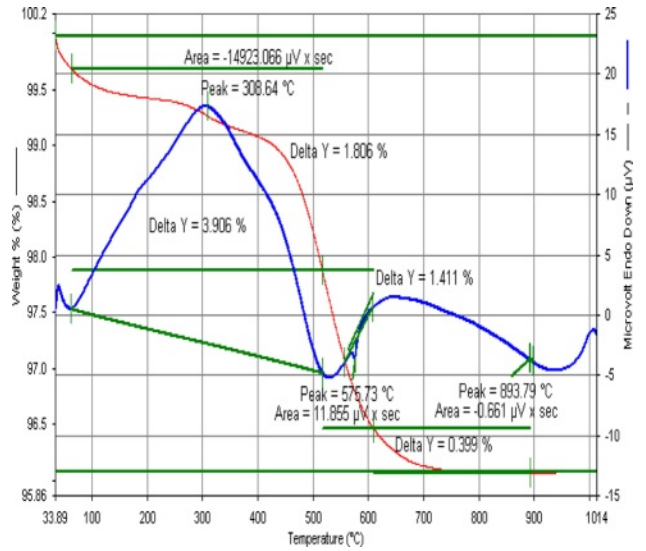


Fig. 14. E10 engobe DTA.

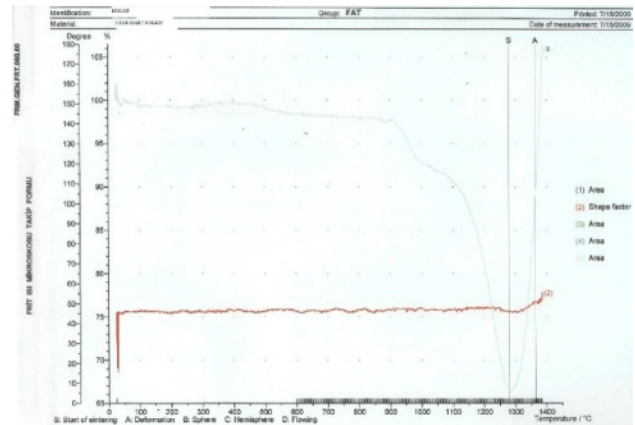


Fig. 15. Thermal microscopy graphic of the standart engobe.

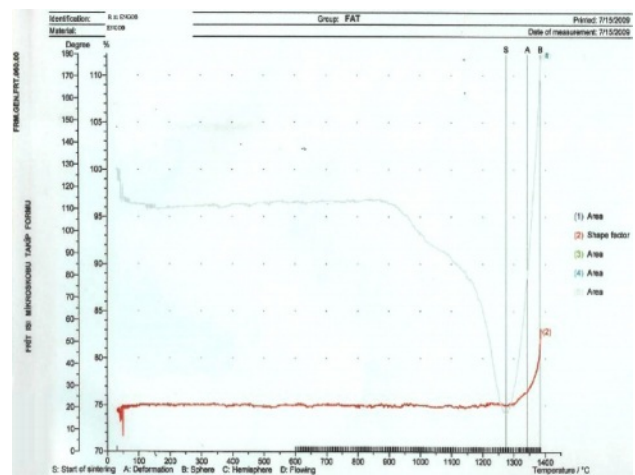


Fig. 16. Thermal microscopy graphic of the E11 engobe.

in Çanakkale Seramik Factory. Thermal expansion analysis of the acquired engobes has been performed with NETZSCH DIL 402 PC dilatometre device to indicate their thermal coefficient and in consequence of controlling the cohesion with the structure to use.

For dilatometre analysis; the engobes which were prepared with wet grinding was dewatered in plaster moulds and sintered in accordance with tile scouring cycle.

An appropriate dimension of dilatometer bar was prepared from sintered engobe mass and placed into

the specimen holder which is parallel with measuring direction. This device keeps the specimen of which piston measurement will be performed suspended by applying adjustable stable force continuously to the edge of the specimen. Through this movement of the piston the change of the specimen size is measured. Dilatometer analysis of the acquired engobe specimen was performed at 550 K' with 25/10.0 (K/min) heating rate. After the analysis of the engobes has been showed that the thermal expansion value is suitable for using with the standart glaze.

According to the heat microscope analysis getting sphere heat for standard engobe could not be observed. Figure 18-20. Standart engobe melting rate for the sphere heat is higher. This shows that sintering temperature of the E11 engobe is lower than standart engobe's. So E11 engobe contains low porosity (Fig. 16).

E11 engobe melting rate for the sphere heat is lower. So the sphere can be observed as shown in Fig. 19.

Sphere heat for E10 engobe could not be observed by heat microscope analysis E10 engobe melting rate for the sphere heat is higher has been shown in Fig. 20.

The sintering and deformation temperatures for the engobes has been shown in Fig. 18, Fig. 19, and Fig. 20. Also, the comparative thermal microscopy graphics of the E10, E11 and the standart engobe have been shown in Fig. 21 [15].

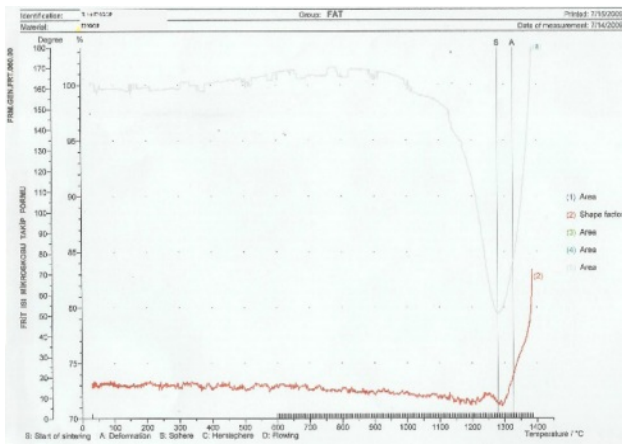


Fig. 17. Thermal microscopy graphic of the E10 engobe.

Characteristic temperatures:		Heating profile:	
Ritening temperature:	1279°C	Soam. Heating rate:	End temp. Dwell time
Deformation temperature:	1384°C	1 40°C/min	1450°C 00:00
Sphere temperature:	n.d.	2 0°C/min	0°C 00:00
Hemisphere temperature:	n.d.	3 0°C/min	0°C 00:00
Flow temperature:	n.d.	4 0°C/min	0°C 00:00
FRIT 51120 (169x4) / RO 540 (169x56-15), excluding sphere and flow temperature		0°C/min 0°C 00:00	
Measurement parameters:		Conditions for new images:	
Images:	161	First image at:	800°C
Measured date count:	2450	Images at least every:	-
Tracked corner angle:	left	Area change:	8%
Operator:	Vahide YILMAZ	Corner angle change:	12%
Device:	EMM2	Shape factor change:	8%
Measured data folder:	C:\EMM2\Date\G0206140\MM0907151	Temperature change:	5°C
Notes:	16.7.2009 FRIT FABRIKASI		

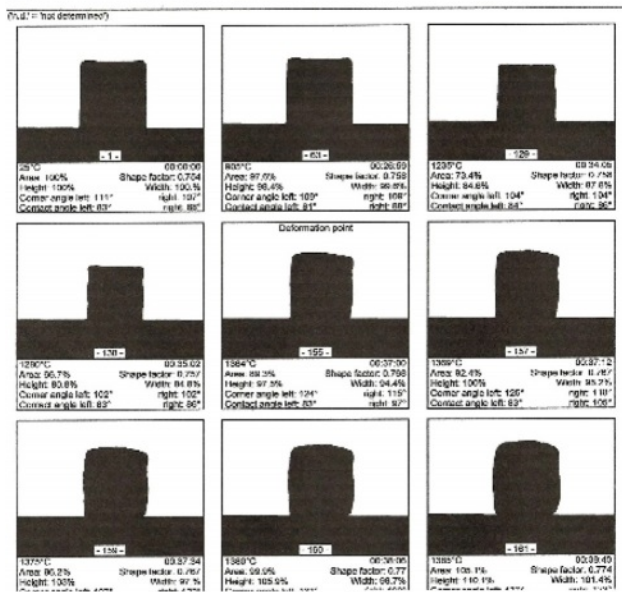


Fig. 18. Thermal microscopy image of the standart engobe.

Characteristic temperatures:		Heating profile:	
Sintering temperature:	1274°C	Soam. Heating rate:	End temp. Dwell time
Deformation temperature:	1343°C	1 40°C/min	1450°C 00:00
Sphere temperature:	1383°C	2 0°C/min	0°C 00:00
Hemisphere temperature:	n.d.	3 0°C/min	0°C 00:00
Flow temperature:	n.d.	4 0°C/min	0°C 00:00
FRIT 51120 (169x4) / RO 540 (169x56-15), excluding sphere and flow temperature		0°C/min 0°C 00:00	
Measurement parameters:		Conditions for new images:	
Images:	162	First image at:	800°C
Measured date count:	2426	Images at least every:	-
Tracked corner angle:	left	Area change:	5%
Operator:	Vahide YILMAZ	Corner angle change:	12%
Device:	EMM2	Shape factor change:	5%
Measured data folder:	C:\EMM2\Date\G0206140\MM0907152	Temperature change:	5°C
Notes:	16.7.2009 FRIT FABRIKASI		

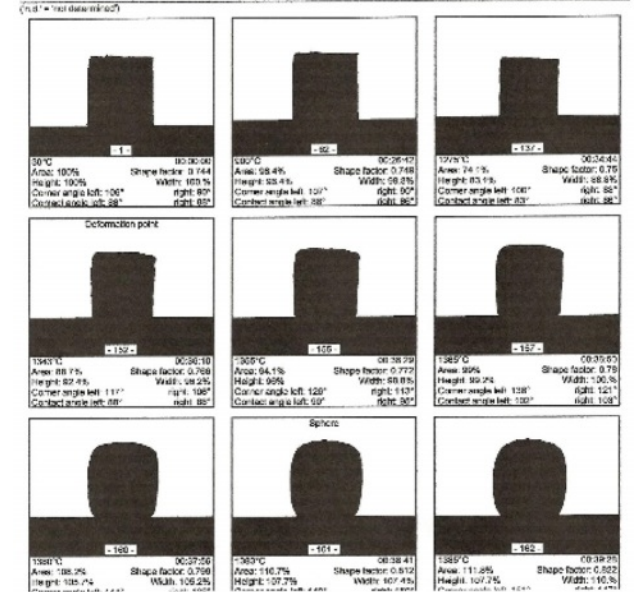


Fig. 19. Thermal microscopy image of the E11 engobe.

Appendix

Seger Formulations have been calculated as below for the engobes.

Table 4. Standard engobe seger formulations.

Seger :	%mol				
CaO	0,031	0,365	2,3515	SiO ₂	55,17
MgO	0,004	0,044	0,284	Al ₂ O ₃	30,56
Na ₂ O	0,038	0,455	2,9314	CaO	1,84
K ₂ O	0,011	0,135	0,8705	MgO	0,16
BaO	0,000	0,000	0,000	Na ₂ O	2,54
ZnO	0,000	0,000	0,000	K ₂ O	1,14
Total	0,084	1,000		BaO	0,00
				ZnO	0,00
SiO ₂	0,893	10,662	68,637	ZrO ₂	4,28
ZrO ₂	0,032	0,386	2,4876	B ₂ O ₃	0,98
B ₂ O ₃	0,013	0,156	1,006		
Al ₂ O ₃	0,279	3,329	21,432		
Total	1,2178	15,584	100		

Table 5. E7 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen A	16,0	1,88	7,39	6,03	0,12	0,02	0,01	0,03	0,05	0,44				
Kil 1	17,0	1,25	9,80	4,88	0,18	0,25	0,05	0,02	0,08	0,42				
Kil 2	0													
Frit	35	2,41	16,74	8,189	0,02	0,04	2,98	0,11	1,09	0,09				1,77
Alümina	7,0	0,02	10,82											
Kuvars	11,0	0,02	10,82											
Zirkon	14,0	0,05	4,47	0,13	0,01								7,96	
Dolomit	0													
Albit	0													
Total	100,0	5,637	49,221	26,164	0,338	0,303	3,050	0,155	1,244	0,952	0	0	7,965	1,772

Table 6. E7 seger formulations.

Seger	%mol				
CaO	0,054	0,615	4,3382	SiO ₂	53,99
MgO	0,004	0,044	0,3094	Al ₂ O ₃	28,70
Na ₂ O	0,020	0,227	1,5986	CaO	3,35
K ₂ O	0,010	0,114	0,8068	MgO	0,17
BaO	0	0	0	Na ₂ O	1,37
ZnO	0	0	0	K ₂ O	1,04
Total	0,089	1,000		BaO	0,00
SiO ₂	0,820	9,263	65,334	ZnO	0,00
ZrO ₂	0,065	0,731	5,157	ZrO ₂	8,74
B ₂ O ₃	0,025	0,287	2,027	B ₂ O ₃	1,94
Al ₂ O ₃	0,257	2,896	20,429		
Total	1,1671	14,178	100		

Table 7. E8 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen A	16,7	1,97	7,71	6,30	0,13	0,02	0,01	0,03	0,05	0,46	0	0	0	0
Kil 1	16,7	1,23	9,63	4,80	0,18	0,24	0,05	0,02	0,08	0,41				
Kil 2	0													
Frit	33,3	2,30	15,93	7,79	0,02	0,04	2,84	0,11	1,04	1,04	0,09		0	1,69
Alümina	7,4	0,02	0	7,33										
Kuvars	11,1	0,02	10,91											
Zirkon	14,8	0,05	4,73	0,14	0,01								8,42	
Dolomit	0													
Albit	0													
Total	100,0	5,584	48,912	26,348	0,339	0,297	2,905	0,151	1,193	0,960	0	0	8,420	1,686

Table 8. E8 seger formulations.

Seger	%mol		
CaO	0,052	0,610	4,1457
MgO	0,004	0,044	0,3017
Na ₂ O	0,019	0,226	1,5382
K ₂ O	0,010	0,120	0,8159
BaO	0	0	0
ZnO	0	0	0
Total	0,085	1	
SiO ₂	0,815	9,579	65,149
ZrO ₂	0,068	0,804	5,4706
B ₂ O ₃	0,024	0,285	1,935
Al ₂ O ₃	0,258	3,035	20,644
Total	1,1662	14,703	100
SiO ₂			53,63
Al ₂ O ₃			28,89
CaO			3,18
MgO			0,17
Na ₂ O			1,31
K ₂ O			1,05
BaO			0
ZnO			0
ZrO ₂			9,23
B ₂ O ₃			1,85

Table 9. E9 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen A	18,2	2,14	8,40	6,86	0,14	0,02	0,01	0,03	0,06					
Kil 1	11,4	0,84	6,57	3,27	0,12	0,17	0,04	0,01	0,05	0,28				
Kil 2	0													
Frit	50	3,45	23,91	11,69	0,03	0,06	4,26	0,16	1,56	0,13				2,53
Alümina	9,1	0,03	0	9,02					0,03					
Kuvars	11,3	0,02	11,11											
Zirkon	0													
Dolomit	0													
Albit	0													
Total	100	6,477	50,002	30,845	0,290	0,241	4,313	0,202	1,700	0,916				2,532

Table 10. E9 seger formulations.

Seger:	%mol				
CaO	0,077	0,646	5,9644	SiO ₂	54,92
MgO	0,005	0,042	0,3911	Al ₂ O ₃	33,88
Na ₂ O	0,027	0,230	2,1229	CaO	4,74
K ₂ O	0,010	0,082	0,7547	MgO	0,22
BaO	0,000	0,000	0	Na ₂ O	1,87
ZnO	0,000	0,000	0	K ₂ O	1,01
Total	0,119	1,000		BaO	0
SiO ₂	0,833	6,989	64,584	ZnO	0
ZrO ₂	0	0	0	ZrO ₂	0
B ₂ O ₃	0,036	0,305	2,816	B ₂ O ₃	2,78
Al ₂ O ₃	0,302	2,536	23,417		
Total	1,1721	10,831	100		

Table 11. E5 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen A	22	2,59	10,16	8,29	0,17	0,02	0,02	0,04	0,07	0,61				
Kil 1	28	2,07	16,15	8,04	0,30	0,41	0,09	0,03	0,13	0,69				
Kil 2	10	1,58	5,36	3,04	0,13	0,08	0,02	0,01	0,05	0,25				
Frit	14	0,97	6,7	3,27	0,01	0,02	1,19	0,04	0,44	0,04				0,71
Kuvars	9	0,01	8,85											
Zirkon	6	0,02	1,92	0,06	0,01								3,41	
Dolomit	4	1,85	0,05	0,04			1,2	0,82	0,03					
Albit	10	0,01	6,83	2,12	0,01	0,03	0,06	0,03	0,90	0,02				
Total	103	9,093	56,013	24,859	0,617	0,561	2,577	0,961	1,608	1,591			3,413	0,709

Table 12. E5 seger formulations.

Seger :	%mol				
CaO	0,046	0,408	3,4651	SiO ₂	60,29
MgO	0,024	0,213	1,8081	Al ₂ O ₃	26,76
Na ₂ O	0,026	0,230	1,9531	CaO	2,77
K ₂ O	0,017	0,150	1,2743	MgO	1,03
BaO	0,000	0,000	0	Na ₂ O	1,73
ZnO	0,000	0,000	0	K ₂ O	1,71
Total	0,113	1,000		BaO	0
				ZnO	0
SiO ₂	0,934	8,264	70,293	ZrO ₂	3,67
ZrO ₂	0,028	0,246	2,0895	B ₂ O ₃	0,76
B ₂ O ₃	0,010	0,090	0,767		
Al ₂ O ₃	0,244	2,159	18,350		
Total	1,2152	11,764	100		

Table 13. E10 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen B	27	3,28	12,65	10,18	0,22	0,02	0,02	0,01	0,03	0,54				
Kil 1	15	1,11	8,65	4,31	0,16	0,22	0,05	0,01	0,07	0,37				
Kil 2														
Frit	17	1,17	8,13	3,98	0,01	0,02	1,45	0,05	0,53	0,05				0,86
Kuvars	22	0,03	2,56	0,07	0,01	0	0	0	0	0				
Zirkon	8	0,03	2,56	0,07	0,01								4,55	
Dolomit														
Albit	13	0,01	8,88	2,76	0,01	0,04	0,07	0,04	1,17	0,02				
Total	102	5,626	62,507	21,298	0,407	0,297	1,596	0,116	1,796	0,973			4,551	0,861

Table 14. E10 seger formulations.

Seger :	%mol					
CaO	0,029	0,403	2,0795	SiO ₂		66,21
MgO	0,003	0,041	0,2119	Al ₂ O ₃		22,56
Na ₂ O	0,029	0,410	2,1132	CaO		1,69
K ₂ O	0,010	0,146	0,755	MgO		0,12
BaO	0,000	0,000	0	Na ₂ O		1,90
ZnO	0,000	0,000	0	K ₂ O		1,03
Total	0,071	1,000		BaO		0
SiO ₂	1,042	14,731	76,005	ZnO		0
ZrO ₂	0,037	0,523	2,6995	ZrO ₂		4,82
B ₂ O ₃	0,012	0,175	0,902	B ₂ O ₃		0,91
Al ₂ O ₃	0,209	2,953	15,234			
Total	1,2999	19,382	100			

Table 15. E11 chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Kaolen B	12,5	1,52	5,86	4,72	0,10	0,01	0,01	0,01	0,01	0,25	0,00	0,00	0,00	0,00
Kil 1	20,0	1,48	11,53	5,74	0,22	0,29	0,06	0,02	0,09	0,49	0,00	0,00	0,00	0,00
Kil 2	5,00	0,79	2,68	1,52	0,07	0,04	0,01	0,00	0,02	0,12	0,00	0,00	0,00	0,00
Frit	22,5	1,55	10,76	5,26	0,01	0,03	1,92	0,07	0,70	0,06	0,00			1,14
Alumina	4,00	0,01	0,00	0,96	0,00	0,00	0,00	0,00	0,01	0,00	0,00	0,00	0,00	0,00
Kuvars	12,00	0,02	11,80											
Zirkon	14,00	0,05	4,47	0,13	0,01								7,96	
Dolomit														
Albit	10,00	0,01	6,83	2,12	0,01	0,03	0,06	0,03	0,90	0,02				
Total	100	5,423	53,942	23,451	0,415	0,399	2,60	0,126	1,741	0,938			7,965	1,39

Table 16. E11 seger formulations.

Seger :	%mol				
CaO	0,037	0,472	2,8559	SiO ₂	58,52
MgO	0,003	0,040	0,2451	Al ₂ O ₃	25,44
Na ₂ O	0,028	0,360	2,1804	CaO	2,23
K ₂ O	0,010	0,128	0,7747	MgO	0,14
BaO	0,000	0,000	0,000	Na ₂ O	1,89
ZnO	0,000	0,000	0	K ₂ O	1,02
Total	0,078	1,000		BaO	0
				ZnO	0
SiO ₂	0,889	11,525	69,797	ZrO ₂	8,64
ZrO ₂	0,065	0,830	5,0271	B ₂ O ₃	1,24
B ₂ O ₃	0,016	0,210	1,270		
Al ₂ O ₃	0,230	2,947	17,850		
Total	1,2101	16,512	100		

Table 17. Frit chemical analysis.

Raw Material	%	Fire Loss	SiO ₂	Al ₂ O ₃	Fe ₂ O ₃	TiO ₂	CaO	MgO	Na ₂ O	K ₂ O	BaO	ZnO	ZrO ₂	B ₂ O ₃
Mermer	16	6,78	0,31	0,17	0,02		8,33	0,23	0,08					
Borik Asit	16,0	0	0	0	0	0	0	0	0	0	0	0	0	9,00
Alumina	2,0	0,01	0,00	1,98	0,00	0,00	0,00	0,00	0,01	0,00	0,00	0,00	0,00	0,00
Kuvars	24,0	0,03	23,60											
Zirkon	9,0	0,03	2,88	0,08	0,01								6,89	
Albit	10,00	0,01	6,83	2,12	0,01	0,03	0,06	0,03	0,90	0,02				
Total	100	6,932	48,368	8,745	0,060	0,05	8,382	0,282	0,863	3,911			6,895	9,000

Table 18. Seger Formulations of frit.

Seger :	%mol				
CaO	0,150	0,705	11,608	SiO ₂	55,91
MgO	0,007	0,033	0,5462	Al ₂ O ₃	10,11
Na ₂ O	0,014	0,066	1,0794	CaO	9,69
K ₂ O	0,042	0,196	3,2263	MgO	0,33
BaO	0,000	0,000	0,000	Na ₂ O	1,00
ZnO	0,000	0,000	0	K ₂ O	4,52
Total	0,212	1,000		BaO	0
				ZnO	0
SiO ₂	0,806	3,798	62,518	ZrO ₂	7,97
ZrO ₂	0,056	0,264	4,3473	B ₂ O ₃	10,40
B ₂ O ₃	0,129	0,609	10,026		
Al ₂ O ₃	0,086	0,404	6,649		
Total	1,0772	6,0752	100		